## **Oxygen-Evolution Catalysts**

## Synergistic Effect between Metal–Nitrogen–Carbon Sheets and NiO Nanoparticles for Enhanced Electrochemical Water-Oxidation Performance\*\*

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Abstract: Identifying effective means to improve the electrochemical performance of oxygen-evolution catalysts represents a significant challenge in several emerging renewable energy technologies. Herein, we consider metal-nitrogen-carbon sheets which are commonly used for catalyzing the oxygenreduction reaction (ORR), as the support to load NiO nanoparticles for the oxygen-evolution reaction (OER). FeNC sheets, as the advanced supports, synergistically promote the NiO nanocatalysts to exhibit superior performance in alkaline media, which is confirmed by experimental observations and density functional theory (DFT) calculations. Our findings show the advantages in considering the support effect for designing highly active, durable, and cost-effective OER electrocatalysts.

he ever-growing public concerns about the energy crisis and energy security call for the development of sustainable energy as viable alternatives to fossil fuels.<sup>[1]</sup> Solar energy has the potential to meet global energy demand for clean and renewable energy, provided its conversion and storage issues are settled.<sup>[2]</sup> The electrochemical splitting of water offers a promising process enabling the conversion from solar to storable hydrogen fuels.<sup>[3]</sup> In this process, the oxygenevolution reaction (OER) is a kinetic barrier, requiring

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a large overpotential that deteriorates the overall efficiency of water splitting.<sup>[4]</sup> At present, ruthenium and iridium oxides are regarded as the benchmarking OER catalysts, but the high cost and low abundance impede their use on a large scale.<sup>[5]</sup> Therefore, the exploration of active, durable, and inexpensive first-row transition-metal oxides (MOs) with commensurate OER performance is highly pursued.<sup>[6]</sup> Among them, nickelbased oxides have been proposed as good candidates and many efforts have focused on identifying means to further improve their performance.<sup>[7]</sup>

Besides tuning the size, shape, composition, and crystallinity of active component of catalysts,<sup>[8]</sup> optimizing the interaction between the active component and the support could boost the catalytic performance. This is because the support could raise the dispersity of the active sites, facilitating the mass-transport and electron-transfer kinetics.<sup>[9]</sup> Incorporating active materials with functional supports has been shown to be an effective means of raising the activity of OER catalysts.<sup>[6e, 7a, 10]</sup> For instance, Dai and co-workers found that the enhanced OER activity of Co<sub>3</sub>O<sub>4</sub> nanoparticles (NPs) was obtained by anchoring them on nitrogen-doped graphene.<sup>[10c]</sup> Other reports also showed that manganese, cobalt, and nickel oxides deposited on noble metals can exhibit superior OER activities in contrast to their support-free counterparts.<sup>[6e, 7a, 10d]</sup> The substantial enhancement is believed to originate from the synergetic effect between the metal oxides and the supports.

Metal-nitrogen-carbon (MNC) materials, in which the metal is either Fe and/or Co, are emerging as the most promising oxygen-reduction catalysts.<sup>[11]</sup> And the active sites are ascribed to central metal ions stabilized by nitrogen functional groups on carbonaceous surfaces, showing distinct interactions with oxygen molecules and the intermediates.<sup>[11,12]</sup> Furthermore, the introduction of heteroatoms tunes the electron-donating/withdrawing capability of the carbon basal plane. These unique properties theoretically suggest that MNC materials are potential catalyst supports. Inspired by these ideas, herein, as a proof-of-concept experiment, we first take NiO as active component to evaluate the support effect of MNC sheets. Impressively, the FeNC sheets/NiO exhibits excellent OER performance including high activity and long term stability.

As presented in Scheme 1, the synthesis begins by coating graphene oxide (GO) sheets with polydopamine (PDA) which can deposit on virtually any surface through the self-polymerization of dopamine in an alkaline environment and can subsequently chelate with many types of metal ions.<sup>[13]</sup> Thus the GO@PDA sheets are impregnated with Fe and Co



**Scheme 1.** Schematic illustration of the synthesis of metal-nitrogencarbon (MNC) sheet/NiO NPs.

precursors which after heat-treatment and acid-leaching give Fe-nitrogen-carbon (FeNC) and Co-nitrogen-carbon (CoNC) sheets, respectively. Metal-free NC sheets are also prepared as the control sample by heat-treating GO@PDA sheets directly. After loading the sheets with NiO NPs to perform OER, the FeNC sheets display the most improved performance which is a result of the low electron-transfer barrier and high electron-coupling capability.

The morphology of GO@PDA sheets was first investigated by transmission electron microscopy (TEM). As shown in Figure 1 a, several thick and wrinkled sheets with smooth surface and sizes on the micrometer scale are observed. And no isolated PDA NPs appear, suggesting the completely



**Figure 1.** TEM images of a) GO@PDA and b,c) FeNC sheets; d) FT-IR spectra of GO@PDA and FeNC sheets; e) HAADF-STEM image of FeNC sheets and the corresponding element-mapping images of f) C, g) N, h) O, and i) Fe; j),k) TEM images of FeNC sheets/NiO; l) XRD patterns of GO@PDA sheets, FeNC sheets, and FeNC sheets/NiO.

selective polymerization of dopamine on the GO surface. When converted into FeNC sheets, though their surfaces becomes rough, the structural integrity of sheet is well preserved (Figure 1b). The highly magnified TEM image reveals that the surface of the FeNC sheets is composed of many mesopores which are derived from the unstable Fe species after acid-leaching (Figure 1c). The porous character will clearly increase the surface area and facilitate gas diffusion, thus boosting the activity of the catalyst. Meanwhile, a few of residual Fe NPs are also observed with a spacing of 0.202 nm which can be assigned to Fe(110) plane. Fourier-transform infrared (FT-IR) spectra illustrate that the characteristic peaks of several functional groups belonging to PDA disappear from the surface of FeNC sheets after heattreatment (Figure 1 d). Meanwhile the thickness of FeNC sheets (ca. 1.5 nm) shrinks in comparison to that of GO@PDA sheets (ca. 3.5 nm), as confirmed by atomic force microscopy (AFM) images (Figure S1 in the Supporting Information). The X-ray diffraction patterns (XRD) display two peaks centered at approximately 25° and 44°, corresponding to carbon (002) and (101) diffractions for FeNC sheets, in contrast with one broad peak for GO@PDA, indicative of the high graphitic degree of the FeNC sheets (Figure 11). Similar TEM, FT-IR, AFM and XRD results are observed for CoNC and NC sheets, except for the relatively smooth surface of NC sheets (Figure S2–S6).

Furthermore, the Brunauer–Emmett–Teller (BET) surface areas of the three samples are estimated by  $N_2$ 

adsorption–desorption isotherms (Figure S7a). The BET surface areas and pore volumes are found to be  $439 \text{ m}^2 \text{g}^{-1}$  and  $0.44 \text{ cm}^3 \text{g}^{-1}$  for FeNC sheets,  $513 \text{ m}^2 \text{g}^{-1}$  and  $1.00 \text{ cm}^3 \text{g}^{-1}$  for CoNC sheets, and  $323 \text{ m}^2 \text{g}^{-1}$  and  $0.20 \text{ cm}^3 \text{g}^{-1}$  for NC sheets, respectively. The pore size distribution reveals that CoNC sheets have more mesopores than FeNC and NC sheets (Figure S7b). Reasonably, NC sheets without porous surface take the lowest BET surface area and pore volume. However, their discrepancies between FeNC and CoNC sheets are inscrutable, which may be explained by their compositions.

To this end, X-ray photoelectron spectroscopy (XPS) was performed on FeNC, CoNC and NC sheets. As shown in Figure S8, the three samples are all composed of C, N, and O. And the characteristic peaks of Fe and Co are also observed for FeNC and CoNC sheets, respectively. These results are further corroborated by energy dispersive spectroscopy (Figure S9). Element mapping images reflect that all the samples have the homogeneous distribution of C, N, and O in their sheet structure. Besides, small amounts of Fe and Co are present in the FeNC and CoNC sheets, respectively (Figure 1e-i and Figure S2-S3). The atomic ratios of Fe and Co are 0.10% and 0.04% (Table S1), consistent with inductively coupled plasma-atomic emission spectroscopy measurements wherein the detected Fe and Co contents are 8892 and 3779 ppm, respectively.

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Additionally, the Raman spectra reveal that FeNC sheets have the lowest  $I_D/I_G$  ratio (Figure S10), implying the relatively low structural defects, which is also verified by the lowest (N + O)/C ratio of FeNC sheets (3.81 %) in contrast to that of CoNC (4.33 %) and NC (8.06 %) sheets (Table S1). Conclusively, the reason for low BET surface area and pore volume of FeNC sheets, in comparison to that of CoNC sheets, may be ascribed to the high residual Fe content and low number of defect sites.

Deposition of NiO NPs on the three supports is easily achieved by the impregnation and pyrolysis process. For a full comparison, graphene (G) sheets were selected as another support and pure NiO NPs with uneven size distribution were obtained by pyrolyzing nickel acetate directly (Figure S11). The NPs with sizes ranging from 5 to 8 nm are fairly well dispersed on FeNC and CoNC sheets (Figure 1j,k and Figure S2h,i). However, a little agglomeration of the NPs happens on NC and G sheets, which may be caused by the smooth surface (Figure S3g and Figure S11b). The highresolution TEM image shows the resolved lattice fringes of NiO(111) plane with a spacing of 0.241 nm. XRD patterns confirm the (111), (200), (220), and (311) diffractions of NiO (PDF#65-2901) as well (Figure 11 and Figure S6).

The OER activities are characterized in a three-electrode electrochemical cell with a rotating disk electrode (RDE) configuration. Cyclic voltammetry (CV) is performed in 0.1M KOH using a scan rate of  $10 \text{ mV s}^{-1}$ , at room temperature. Potentials are converted to the reversible hydrogen electrode (RHE) scale by calibration (Figure S12). Figure 2 a shows that

the FeNC sheets/NiO displays greater current density and a more negative OER onset potential (1.47 V vs. RHE) than the CoNC sheets/NiO (1.49 V) and NC sheets/NiO (1.51 V). In addition, the overpotential required at the current density of  $10 \text{ mA cm}^{-2}$  is always regarded as the benchmark to estimate the activity of OER catalyst. The FeNC sheets/NiO and CoNC sheets/NiO reach this value with a potential of 1.62 and 1.64 V, respectively. Of note, G sheets/NiO and pure NiO show weak OER activities and the supports have little activity contribution. Figure 2b reveals that Tafel slope of FeNC sheets/NiO is approximately 76 mV/decade, which is smaller than that of CoNC sheets/NiO and NC sheets/NiO, indicating the higher OER kinetics. The surface concentration of redox active Ni centers on the electrode can be extracted from the slope of the linear relationship between the peak current of the reduction wave and scan rate.<sup>[14]</sup> Therefore, the coverage of redox active Ni centers is calculated to be 51.94, 60.52, and 40.55 nmol cm<sup>-2</sup> for FeNC sheets/NiO, CoNC sheets/NiO and NC sheets/NiO, respectively (Figure S13). In view of the fact that OER occurs only on the surface of the catalyst, these values represent the upper limits to the real number of catalytic sites. Thus, the lower turnover frequency (TOF) limit associated with FeNC sheets/NiO at  $\eta = 350 \text{ mV}$  is  $0.2 \text{ s}^{-1}$ , which is higher than that of CoNC sheets/NiO  $(9.5 \times 10^{-2} \text{ s}^{-1})$ and NC sheets/NiO ( $8.5 \times 10^{-2} \text{ s}^{-1}$ ), highlighting the most positive effect of FeNC sheets in promoting OER activity. Based on these results, the FeNC sheets/NiO exerts high OER activity which is comparable to commercial RuO<sub>2</sub> (Figure S14) and many non-precious-metal-based catalysts

E (V vs. Ag/AgCI) 0.0 20 <del>-</del> b а 0.1 0.2 0.3 0.4 0.5 0.7 FeNC sheets/NiO FeNC sheets/NiO 1.65 CoNC sheets/NiO CoNC sheets/NiC 16 NC sheets/NiO NC sheets/NiC 1.62 G sheets/NiO NiO E (V vs. RHE) (mA cm<sup>-2</sup>) FeNC sheets 1.59 CoNC sheets NC sheets 1.56 G sheets 1.53 1.50 1.0 1.1 1.2 1.3 1.5 1.6 1.4 1.7 10 С E (V vs. RHE) *j* (mA cm<sup>-2</sup>) d 30 FeNC sheets/NiC 70 CoNC sheets/NiO 25 60 NC sheets/NiO 50 20 Im(Z) / Ohm *j* (mA cm<sup>-2</sup>) 40 15 30 20 10 10 5 0 0 † 0 40 80 140 160 60 100 120 180 200 6000 10000 2000 4000 8000 Re(Z) / Ohm Time (s)

**Figure 2.** a) CV curves for OER on FeNC, CoNC, NC, and G sheets with and without NiO NPs at a scan rate of  $10 \text{ mVs}^{-1}$ ; b) the corresponding Tafel plots; c) the electrochemical impedance plots obtained at a potential of 1.62 V vs. RHE; and d) a chronoamperometric test for OER over the FeNC sheets/NiO catalyst at 1.62 V vs. RHE. All the measurements are performed in 0.1 M KOH at room temperature with RDE rotation speed of 1000 rpm. Catalyst loading: approximately 0.24 mg cm<sup>-2</sup>.

(Table S2).<sup>[6a,b,7f,10b,c,14,15]</sup>

The electrochemical impedance spectroscopy (EIS) technique is used to provide deep insight into the kinetics during OER process. As shown in Figure 2c, a lowest charge-transfer resistance is observed for FeNC sheets/NiO, associated with its superior OER activity. As the supports, CoNC and NC sheets have more defect sites, resulting in higher charge-transfer barrier at OER condition and thus inferior activities of the catalysts. Consequently, the good electron-transfer kinetics is a prime requirement for electrocatalyst to exhibit high activity. To assess the OER stability of FeNC sheets/NiO, a chronoamperometric test is carried out. Figure 2d reveals that a strong activity maintenance over 10000 s is obtained at approximately  $10 \text{ mA cm}^{-2}$  with the applied potential of 1.62 V, suggesting the long-term durability.

The interaction between the supports and NiO NPs is analyzed by XPS spectra, in which the

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Figure 3. Ni  $2p_{3/2}$  region of XPS spectra for NiO NPs supported on FeNC, CoNC, and NC sheets.

Ni $2p_{3/2}$  region can be deconvoluted into two peaks (Figure 3).<sup>[16]</sup> For NC sheets/NiO, the binding energies are located at 853.4 and 854.9 eV, which shift to higher values for CoNC sheets/NiO (853.6 and 855.4 eV) and FeNC sheets/NiO (854 and 855.7 eV). The X-ray absorption near-edge structure (XANES) spectra of these samples show the Ni K-edge of the FeNC sheets/NiO shifts to relatively higher energy, indicating the increased oxidation state of Ni atoms (Figure S15).<sup>[17]</sup> These phenomena demonstrate that FeNC sheets have the stronger electron coupling effect that facilitates electron transfer from NiO NPs to FeNC sheets. Previous work has indicated that the formation of the O–O bond in OOH adsorbate (Step 3, Figure 4) and the deproto-



*Figure 4.* Schematic illustration of the OER pathways. The colored numbers are the kinetic energy barriers.

nation of the oxyhydroxide group (Step 4, Figure 4) determine the electrocatalytic efficiency of OER.<sup>[4a]</sup> For the FeNC sheets/NiO, the NiO NPs, as electron donors, become more electrophilic, facilitating the nucleophilic reaction by OH<sup>-</sup> with the intermediates and thus accelerating OER. Moreover, according to the Lewis acid–base concept, the electron donation from NiO NPs to FeNC sheets makes the NiO NPs more acidic, which will benefit the interaction with OH<sup>-</sup>, leading to the improved OER activity. These results highlight the synergetic effect between FeNC sheets and NiO NPs.

To confirm the experimental observations, density functional theory (DFT) calculation is employed to provide deeper insights into the detailed OER pathways (see calculation details in Supporting Information). As shown in Figure 4, FeNC/NiO and CoNC/NiO display no obvious discrepancy in the kinetic energy barriers for Step 2 and 3. However, the energy barrier of the rate determining step (Step 4) on FeNC/NiO is 3.23 eV, which is significantly lower than that of CoNC/NiO (3.41 eV). NC/NiO has the highest energy barriers for Steps 2–4, indicative of its unfavorable OER kinetics. Overall, these results are in agreement with the experimental observations, validating the speculations from XPS and XANES results and the superior promoting effect of the FeNC sheets.

To evaluate the Faradaic yield for OER, the FeNC sheets/ NiO catalyst is loaded onto carbon paper (CP) and tested at a constant potential. Synchronously, the evolved gas during the reaction is monitored volumetrically using a gas burette at room temperature and atmospheric pressure. As shown in Figure S16, a 97% Faradaic yield is estimated based on the observed gas volume and the theoretical gas volume calculated by the charge passed through the electrode. The 3% loss may be ascribed to the dissolved gas in the electrolyte and the attached bubbles on electrode surface. Mass spectrum reveals that the evolved gas is composed of only O<sub>2</sub> and H<sub>2</sub>. And a 22 h chronoamperometric test on the CP electrode shows no activity decrease, confirming the stability of the FeNC sheets/ NiO catalyst (Figure S17).

In summary, we have investigated MNC sheets as the supports for NiO nanocatalysts for the OER, in which the FeNC sheets promote the catalysts to exhibit both high activity and long-term stability in alkaline medium. The experimental observations and DFT calculations confirm that the activity improvement originates from low electron-transfer barrier and high electron-coupling capability of the FeNC sheets, highlighting the synergistic effect between the FeNC sheets and NiO NPs. Our findings show the support effect should be considered when designing advanced and cost-effective OER electrocatalysts, which may find applications in energy conversion technologies including not only alkaline water electrolyzers but also metal–air batteries.

**Keywords:** electrocatalysis · metal-nitrogen-carbon sheets · nanoparticles · oxygen-evolution reaction · synergistic effect

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